Photoluminescence Study of Erbium-Mixed Alkylated Silicon Nanocrystals

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Abstract—Alkylated silicon nanocrystals (C11-SiNCs) were prepared successfully by galvanostatic etching of p-Si(100) wafers followed by a thermal hydrosilation reaction of 1 -undecene in refluxing toluene in order to extract C11-SiNCs from porous silicon. Erbium trichloride was added to alkylated SiNCs using a simple mixing chemical route. To the best of our knowledge, this is the first investigation on mixing SiNCs with erbium ions (III) by this chemical method. The chemical characterization of C11-SiNCs and their mixtures with Er3+ (Er/C11-SiNCs) were carried out using X-ray photoemission spectroscopy (XPS). The optical properties of C11-SiNCs and their mixtures with Er3+ were investigated using Raman spectroscopy and photoluminescence (PL). The erbium mixed alkylated SiNCs shows an orange PL emission peak at around 595 nm that originates from radiative recombination of Si. Er/C11-SiNCs mixture also exhibits a weak PL emission peak at 1536 nm that originates from the intra-4f transition in erbium ions (Er3+). The PL peak of Si in Er/C11-SiNCs mixture is increased in the intensity up to three times as compared to pure C11-SiNCs. The collected data suggest that this chemical mixing route leads instead to a transfer of energy from erbium ions to alkylated SiNCs.

Keywords—Photoluminescence, Silicon Nanocrystals, Erbium, Raman Spectroscopy.

I. INTRODUCTION

The major component in microelectronic and optical communication system is silicon. There are three limitations that are associated with silicon related to their use in optical communication system i.e. Si is an indirect band gap, strong non radiative recombination pathways resulting in a very short non radiative lifetime and there is a mismatch between the band edge luminescence at 1.1 μm (1.1 eV) and the wavelength of 1.55 μm (0.8 eV) that is a requirement for compatibility with optical communication system [1]. This number of factors makes this element a poor light emitter.

The limitation of the spectral mismatch is overcome by incorporating silicon with rare earth ions such as Er, Yb, Nd and Tm. Rare earth ions emission has been widely investigated in silicon last two decades [1]-[3]. Erbium ions especially have played a crucial role in the improvement of optical communication system, due to its luminescence band being at 1.54 μm, which is the requirement in this technology. Erbium is 11th element in the series of rare earth elements which located in the sixth row of the periodic table and the electronic configuration of Er is [Xe]4f14 6s2. It is known that Er presents intense narrow luminescence bands in both visible and near infrared regions [4]. The erbium has two oxidation states i.e. Er2+ and Er3+, out of which the former one is very commonly seen in semiconductors; due to its radiative transition at around 1.54 μm which corresponds to the most interesting wavelength in photonics community. Losing one electron of 4f orbital and both of 6s orbital produces the trivalent erbium (Er3+); thus in this ion the incomplete 4f orbital is shielded by 5s and 5p orbitals resulting in luminescence dependent host. The radiative transition of Er3+ in solid hosts looks like the free ion but with some changes due to Stark splitting [5].

The labels of the energy levels in rare earth ions e.g. 4I15/2 or 4S5/2 correspond to their angular momentum and spin quantum numbers [5]. The letters here are attributed to the total orbital angular momentum of the ion. The total orbital angular momentum is resulted by adding the orbital angular momenta of the individual electrons in the ion following the Clebsch-Gordan series [6]. Thus, the letter S indicates the orbital angular momentum (L) of 0, P of 1, D of 2, F of 3 and etc. Therefore, the letter I presents an L of 6. The superscript number 4 denotes the possible orientation of the total spin angular momentum of the ion, which is given as 2S+1, where S is attributed to quantum numbers of spin. While the subscript presents the total angular momentum of the ion. Consequently, in rare earth ions, it should be considered that each discrete energy level is attributed to 2S+1/2L[J][5].

The emission from trivalent erbium at 1554 nm (due to 4I15/2→4I13/2 transition in erbium) is a standard wavelength in the optical telecommunication system; thus it is important to achieve maximum enhancement in Er emission intensity i.e. long luminescent lifetime and high active concentration of erbium [7]. Recently, the investigation on enhancing the optical activity of erbium ions has been extensively reported [1], [8], [9]. The erbium emission can be increased by i) adding clusters which act as sensitizers for erbium excitation, ii) increasing the fraction of trivalent erbium by changing the local atomic environment and iii) optimizing the rate of erbium excitation by applying local field enhancement in presence of metallic particles [8]. The sensitization effect has recently become attractive subject because of the successful enhancement in erbium emission via range of sensitizers (host materials) including SiNCs, SiO2 and Yb ions [7], [9]-[11].

Among the host materials, Si has received particular interest since the discovery of 1.54 μm light emission in erbium doped silicon at 20 K in 1983 by Ennen and co-workers [12] and...
later they observed electroluminescence at 77 K in 1985 [13]. This effective light emission from silicon-based erbium (with sufficient high concentrations of Er) has led to a novel Si-based optoelectronics and light emitting diodes (LEDs) [14]. Taking into account the importance of 1.54 μm light emission from erbium doped silicon in optical telecommunication system, the mechanism of the energy transfer from Si to erbium has been explained as follow [1]. Generally, the mechanism of the erbium emission of erbium doped silicon is described as followed [1]. Firstly, exciton (electron-hole pair) is generated inside SiNCs by using an excitation source. Then, the exciton recombinates radiatively, which causes photon emission as the energy of the resulted photon depends on the size of Si nanocrystal. A part of radiative recombination energy of the exciton of SiNCs is transferred to 4f shell in trivalent erbium; thus exciting Er³⁺. The emission of erbium at 1.54 μm then occurs as a result of the transition from 1I₁₄/₂ to the ground state 1I₁₅/₂. It is also should be noted that the amount of the energy transferred to Er³⁺ depends strongly on the concentration of erbium in the host material and the distance between Si nanocrystals and trivalent erbium as observed by [9], [15]-[17].

There are several successful techniques that used to incorporate and to produce optically active erbium ions into SiNCs including molecular beam epitaxy [18], sputtering [16], chemical vapor deposition [15], thermal diffusion[19] and ion implantation [20].

Fujiwara et al. [21], [22] observed that Er-doped SiNCs films exhibit two emission bands i.e. 0.81 and 1.54 μm that associated to radiative recombination inside Si nanocrystals and to intra-4f transition (1I₁₄/₂→1I₁₅/₂) in erbium ions, respectively. They studied the correlation between the intensities of the two-luminescence bands as a function of erbium concentration. They found that the intensity of erbium emission band at 1.54 μm increased as the concentration of erbium increased, whereas the intensity of 0.81 μm decreased significantly. They suggested that the energy is transferred from Si nanocrystals to erbium ions. In contrast, [15], [16] found that the intensity of the emission peak of erbium increased with decreasing the erbium concentration. This provides also a clear evidence for the energy transfer from silicon to erbium. Also, they suggested that the intensity of this peak mainly depends on the chemical composition of the matrix where SiNCs are embedded in. However, Fujiwara et al. [21], [22] and Cerqueira et al. [15], [16] both observed that the PL peak of erbium ions was also more intense with small size of SiNCs embedded in silicon matrix.

John et al. [23] and Ji et al. [24] implanted Er³⁺ alongside SiNCs into SiO₂ films, where the SiNCs play as a sensitizer which absorb more photons efficiently. Thus, this leads to a significant improvement in the emission properties of Er ions as the erbium ions covalently bonded with SiNCs. In their work, Er-doped SiNCs samples exhibit NIR light emission, which created by transfer energy from Si nanocrystal to erbium. John and co-workers [23] prepared erbium doped SiNCs by the co-pyrolysis of disilane where Er(mhd), (mhd = 2,2,6,6-tetramethyl1-3,5-heptanedionato) was used as a source of erbium ions. Transmission electron microscopy, selected area electron diffraction, photoluminescence and UV-Vis absorption spectroscopies have been used to characterize SiNCs. X-ray energy dispersive spectroscopy was used to confirm the presence of erbium. In addition, [24] synthesised SiNCs by co-pyrolysis of disilane where erbium ions derived from erbium amindane in a dilute He flow at 1000°C. High resolution electron microscopy, selected area electron diffraction, energy dispersive X-ray and PL spectroscopy were used to characterize the samples. These two methods of preparation erbium doped SiNCs provide high concentrations of erbium centered in given SiNCs, thus improvement in the light emission from Er³⁺ has been achieved. Furthermore, they found that the structural and optical properties of SiNCs depend on the location of the erbium ions, where Er³⁺ directly excited by excitons if they dispersed over the Si nanocrystal.

Kik et al. [25] produced SiNCs in silica using silicon ion implantation followed by high temperature annealing. Erbium ions then implanted into SiNCs doped SiO₂ layer. They found that the energy transfer from silicon to erbium ions by the mechanism of electron-hole pair radiative recombination inside SiNCs. In addition, they observed that electron-hole pairs excite erbium ions with efficiency larger than 55% and the maximum number of erbium ions around Si nanocrystals is near unity.

In this letter, we use a new chemical route to mix alkylated silicon nanocrystals (-SiNCs) with Er³⁺ by dispersing Si nanocrystals in deionized water. Alkylated Si nanocrystals then mixed with aqueous erbium tri chloride. Consequently, the chemical changes in alkylated SiNCs, which caused by Er³⁺ have been investigated using X-ray photoemission spectroscopy (XPS) and Fourier transform infrared spectroscopy (FTIR). Meanwhile, the light emission of Er and the optical properties of C₁₁-SiNCs with their mixtures with Er³⁺ have been studied using UV-Vis spectroscopy, Raman spectroscopy and photoluminescence spectroscopy (PL). To the best of our knowledge, there have been no reports on mixing SiNCs with erbium ions by this chemical route.

II. EXPERIMENTAL

A modified version of the method presented by [26] was used to prepare alkyl-capped (11 carbon atoms) SiNCs. This method [27] forms C₁₁-SiNCs with the same optical characteristics with those explained in [26]. Porous silicon was formed by anodic etching of silicon wafer in HF/EtOH. Alkylated silicon nanocrystals were extracted from the porous silicon by a thermal hydrosilation reaction in refluxing toluene. A silicon chip 1×1 cm² was cut from a silicon wafer (boron-doped p-Si <100>, 10 Ω cm resistivity, Compart Technology, Peterborough, UK) using a diamond scribe. The silicon chip was then placed in electrochemical cell polytetrafluoroethylene (PTFE) with a diameter of 1 cm and using Viton™ O-ring at the base to seal against the silicon chip. In the cell, a total of 2 mL ethanol and hydrofluoric acid HF (1:1 V/V) solution was added. The electrochemical cell was linked to a power supply and control system (Keithley
Four porous silicon chips were then refluxed in 25 mL of dry toluene solution (previously distilled over Na, Merck) which contained 0.4 mL of the alkene (1-undecene \( \text{C}_{11}\text{H}_{22}, \) Merck) for four hours in a Schlenk flask (16 cm in length and 5 cm in diameter). The flask was placed on the magnetic stirrer hot plate linked to a refluxing condenser. During the refluxing time, a clear yellow liquid formed and a flow of nitrogen gas was present in order to keep air out of the Schlenk flask. The boiling point of the solution was around 110°C. Under the UV lamp (\( \lambda = 365 \) nm) an orange color luminescence was emitted by the resulting suspension. After 110°C, under the UV lamp (\( \lambda = 365 \) nm) an orange color luminescence was emitted by the resulting suspension. After 110°C, under the UV lamp (\( \lambda = 365 \) nm) an orange color luminescence was emitted by the resulting suspension. After 110°C, under the UV lamp (\( \lambda = 365 \) nm) an orange color luminescence was emitted by the resulting suspension. After 110°C, under the UV lamp (\( \lambda = 365 \) nm) an orange color luminescence was emitted by the resulting suspension. After 110°C, under the UV lamp (\( \lambda = 365 \) nm) an orange color luminescence was emitted by the resulting suspension. After 110°C, under the UV lamp (\( \lambda = 365 \) nm) an orange color luminescence was emitted by the resulting suspension. After 110°C, under the UV lamp (\( \lambda = 365 \) nm) an orange color luminescence was emitted by the resulting suspension.

The Er/C11-SiNCs mixture was prepared by drop-coating 100 µL of the suspension on a glass slide coverslip using a micropipette (Eppendorf). The optical properties of alkylated SiNCs, their mixtures with erbium ion and erbium trichloride solution were characterized using UV-Vis spectroscopy (Varian, Cary 100 BIO) at Newcastle University. The wavelength range between 200 to 800 nm was used for quartz cuvette of 1 cm path length. This measurement was carried out at room temperature. The samples i.e. C11-SiNCs and their mixtures with erbium trichloride was prepared by adding 4 mL of suspension in the quartz cuvette.

The photoluminescence properties of C11-SiNCs and their mixtures with erbium trichloride were investigated on a Varian 800 Scimitar FTIR over the range of 700 - 4000 cm\(^{-1}\). A thick film of the C11-SiNCs that contain erbium ions was prepared by a drop cast method on silicon substrate using a micropipette (Eppendorf) and left overnight to dry in air at room temperature.

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1-undecene have a diameter of 2.5 nm, which associate to the silicon core of the nanoparticles.

Fig. 1 (a) shows the Si2p spectrum of C11-SiNCs suspension in deionized water dried on gold substrate. The Si2p peak has been fitted with a mixed singlet. This figure presents a peak at a binding energy of 101.3 eV. This binding energy of Si2p at 101.3 eV is consistent for alkylated SiNCs [27]. This peak assigned to a multicomponent peak that contain different species i.e. bulk silicon, Si-C, Si-H and Si-O [29]. The IR spectrum of the C11-SiNCs [26] confirms the presence of Si-C and Si-O. This peak in this figure is similar to that observed by [27], [29]. They determined a peak for alkylated silicon nanocrystals at 101.6 eV. Obviously, the difference between these two binding energies is 0.3 eV. This difference in the binding energy values are predictable because the thickness of the C11-SiNCs film change in every deposition, thus would impact the degree of charging in the film. The charge effect arises from the long alkyl chain that passivates the surface of SiNCs. Also, the charge effect would likely to be due to the agglomeration of SiNCs as clusters rather than diffuse as individual particles as observed by [27], [29].

Fig. 1 (b) shows the O1s core level of C11-SiNCs dried on gold substrate. This peak has been fitted using a mixed singlet. The binding energy at 533.4 eV is assigned to Si-O [31]. Fig. 1 (c) presents a survey spectrum of alkylated silicon nanocrystals. The core lines i.e. Si2p, C1s and O1s are clearly visible in the survey spectrum. Thus, in the film of the investigated sample silicon, carbon and oxygen are present. Some hydrogen atoms may be there in the film but X-ray photoemission spectroscopy cannot detect them [29]. This survey spectrum confirmed also the previous study of IR spectroscopy [26] which exhibits the existence of Si-C and Si-O bonds. The Si-C bonds are anchoring the alkyl chains to dots by covalent bonds while the Si-O bonds arise from the oxidation of unalkylated Si-H [26]. Altogether, the data suggest that the alkyl chains and sub oxides are present on the surface of silicon nanocrystals.

Fig. 2 presents the FTIR spectrum of Er/C11-SiNCs. It can be seen that there are several IR bands of erbium mixed alkylated silicon nanocrystals sample. The IR peak at 2954 cm⁻¹ corresponds to CH₃ stretching mode [26]. The IR peaks that appear at 2921 cm⁻¹ and 2854 cm⁻¹ are assigned to the methylene C-H asymmetric and symmetric stretching mode, respectively [26]. These three IR bands at 2954 cm⁻¹, 2921 cm⁻¹ and 2854 cm⁻¹ prove that the long alkyl chains are present in the film of Er/C11-SiNCs. The band at 1459 cm⁻¹ corresponds to CH₂ bending mode [26], [32]. The two IR bands at 1259 cm⁻¹ and 1220 cm⁻¹ are corresponding to Si-CH₂-R bending mode [26], [33]. Broad IR peak occurs at 1053 cm⁻¹ as a result of gradual oxidation of the investigated sample which correspond to Si-O-Si symmetric and asymmetric stretching modes [26]. In addition, SiO bending vibration modes appear at the region 700 – 900 cm⁻¹ [26]. The IR peaks at 800 cm⁻¹ and 615 cm⁻¹ are attributed to silanol sites and unalkylated Si-H bending vibration mode, respectively [26].

The FTIR spectrum of Er/C11-SiNCs in Fig. 2 is similar to that observed by [26]. There is no considerable change in the vibration modes of Er/C11-SiNCs when compared with that of C11-SiNCs as presented in Table I. Thus, these data suggest that the erbium ions (Er³⁺) do not have effect on the surface of alkylated SiNCs. However, there is no peak determined for erbium trichloride by FTIR spectrum as the erbium does not vibrate in IR range[34]. Thus, XPS is used to determine the surface chemical composition of Er/C11-SiNCs.
TABLE I
IR VIBRATION BANDS THAT OBSERVED IN FTIR SPECTRA FOR PURE SILICON NANOCRYSTALS AND THEIR MIXTURES WITH ERBIUM IONS

<table>
<thead>
<tr>
<th>Functional group</th>
<th>Vibration modes</th>
<th>C11-SiNCs (cm⁻¹)</th>
<th>Er/C11-SiNCs (cm⁻¹)</th>
</tr>
</thead>
<tbody>
<tr>
<td>CH₃ Stretching</td>
<td>2956</td>
<td>2954</td>
<td></td>
</tr>
<tr>
<td>CH Stretching</td>
<td>2926</td>
<td>2921</td>
<td></td>
</tr>
<tr>
<td>CH Stretching</td>
<td>2854</td>
<td>2854</td>
<td></td>
</tr>
<tr>
<td>CH₂ Bending</td>
<td>1459</td>
<td>1459</td>
<td></td>
</tr>
<tr>
<td>SiCH₂R Bending</td>
<td>1259 &amp; 1220</td>
<td>1259 &amp; 1220</td>
<td></td>
</tr>
<tr>
<td>SiOSi Stretching</td>
<td>1050</td>
<td>1053</td>
<td></td>
</tr>
<tr>
<td>SiO Bending</td>
<td>700-900</td>
<td>700-900</td>
<td></td>
</tr>
<tr>
<td>Si(CH₃)₃OH Bending</td>
<td>800</td>
<td>800</td>
<td></td>
</tr>
<tr>
<td>SiH Bending</td>
<td>615</td>
<td>615</td>
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</tr>
</tbody>
</table>

Fig. 2 FTIR spectrum of Er mixed C11-SiNCs deposited on a single crystal silicon wafer.

Fig. 3 (a) presents the XPS spectrum of Si₂p of Er/C₁₁-SiNCs solution dried on gold substrate. This spectrum is similar to that observed by [27]. This peak consists of broad peak with a tail to higher binding energy. This peak has been fitting using mixed singlets. This spectrum is fitted with three peaks with binding energies of 98.6 eV, 102.4 eV and 106.6 eV. They are assigned to different species of Si that are present in Er/C₁₁-SiNCs. The peak at 98.6 eV is assigned to Si [31]. While the peak at 102.4 eV can be attributed to SiO₂, SiₓCᵧ and SiₓOᵧ [29], [35], [36]. Furthermore, the higher binding energy component, at 106.6 eV, can be associated to saturated peak of silicon [29].

Fig. 3 (b) presents the O₁s core level spectrum of Er/C₁₁-SiNCs. This spectrum is fitted using mixed singlets into three components i.e. 530.8 eV, 532.6 eV and 533.4 eV. The binding energies at 530.8 eV assigned to Er₂O₃ [35]-[37]. The binding energy at 532.6 eV corresponds to SiO₂ while the binding energy at 533.4 eV corresponds to SiOₓ [35], [36].

Fig. 3 (c) shows the XPS spectrum of Er₄d for Er/C₁₁-SiNCs. This peak has also been fitted using mixed singlets. The two peaks constitute a doublet with spin orbit splitting of 2 eV that is consistent with [37]. The binding energy at 168.4 eV and 170.4 eV of Er₄d assigned to 4dₓ² and 4dᵧ² of Er₂O₃ [37].

A survey spectrum of erbium mixed alkylated silicon nanocrystals is plotted in Fig. 3 (e). The core lines of Si₂p, O₁s, Er₄d, and C₁s are visible and clear which supports the FTIR study (see Fig. 2), that indicate the presence of both SiO (originated from the oxidation of unalkylated Si-H) and Si-C (bonded the alkyl chains to the dots). Thus, the XPS data confirm the presence of the components of Si-C and Si-O in binding energy at 102.4 eV. In addition, the data in Fig. 3 (e) show that the silicon nanocrystals are surrounded with a combination of surface SiO₂, Er₂O₃ and alkyl chains.

Comparing the two investigated samples i.e. C₁₁-SiNCs and Er/C₁₁-SiNCs, it can be seen that the core level Si₂p for alkylated SiNCs shows one multicomponent peak, see Fig. 1 (a), while the core level Si₂p for erbium mixed alkylated SiNCs presents two peaks, see Fig. 3 (a). It can be explained as result of presence of erbium sesquioxide on the surface of alkylated SiNCs which affect the oxidation states of SiNCs on the surface. Fig. 3 (c) shows the core level Er₄d for Er/C₁₁-SiNCs, which confirm that the erbium trichloride hydrolysis to Er₂O₃ when mixed with alkylated SiNCs. Meanwhile, Fig. 3 (b) presents O₁s of Er/C₁₁-SiNCs that prove that Er₂O₃ is present on the surface of C₁₁-SiNCs. This observation can be also interpreted as a result of exposing C₁₁-SiNCs for oxygen and H₂O for long times which increase the amount of SiₓOᵧ species on the C₁₁-SiNCs surface. These data suggest that the main effect of erbium ions (III) is affecting the oxidation states of alkylated SiNCs.

Fig. 3 (d) presents the XPS spectrum of C₁s for Er/C₁₁-SiNCs. The peak has also been fitting using mixed singlets. This spectrum consist of three peaks i.e. 284.6 eV, 285.8 eV and 288.9 eV which they are correspond to adventitious carbon [38], C-C [39] and Si-C [40], respectively. The FTIR data confirm the presence of Si-C (see Table I).
present a broad absorption tail with a steep rise feature at around 350 nm which can be assigned to the direct band gap transition (ca. 3.4 eV in bulk silicon). It is known that bulk silicon exhibits direct band gap of 3.4 eV and an indirect band gap of 1.1 eV [26]. Therefore, there is a rapid increase in the absorbance with decreasing the wavelength from the wavelength of 600 nm (2.06 eV) that presents the absorption band edge of the indirect band gap of SiNCs [26], [41], [42].

The result shows the absorption increases as the direct gap is approached as observed by [26]. Fig. 4 indicates also that the presence of erbium ions with C11-SiNCs does not change the absorption spectra of silicon nanocrystals. These absorption spectra of Er/C11-SiNCs is similar to those observed by [23] as this manner is constant for silicon nanocrystals due to indirect band gap.

Figs. 5 (a)-(b) display a reflected light image and luminescence image of alkylated SiNCs do not contain erbium trichloride drop cast on glass coverslip. The confocal luminescence images of dried C11-SiNCs present bright features, which are due to the aggregates of C11-SiNCs as shown in Fig. 5 (b). Figs. 5 (c)-(d) present a reflected light image and luminescence image of alkylated SiNCs do contain erbium ions drop cast on glass coverslip. The bright features confirm the presence of clusters of C11-SiNCs, which are associated to Er3+, see Fig. 5 (d). Mainly, the drying process during sample preparation for the microscopy is the original reason for these aggregates (clusters), which determined in Figs. 5 (b), (d) as these clusters were observed by previous works on confocal Raman spectroscopy [27], [43], [44].

These bright regions of the confocal luminescence images were gathered together in order to provide complete average Raman luminescence spectra for C11-SiNCs and Er/C11-SiNCs as presented in Fig. 6. Fig. 6 (a) provides an average Raman spectrum of C11-SiNCs that exhibit the consistent Raman bands of alkylated SiNCs that prepared by this method. The Raman peak at 515 cm⁻¹ is assigned to the first order of crystalline SiNCs as this peak constant for SiNCs [27]. This peak is less than the peak of bulk silicon (520 cm⁻¹) due to quantum confinement effect [27]. The feature at 960 cm⁻¹ can be attributed to the second order of SiNCs [28]. There is also a broad and large peak with a maximum at 4915 cm⁻¹ (642 nm) which attributed to the orange luminescence signal of alkylated SiNCs due to quantum confinement effect [27], [28], [44]. This orange luminescence peak is characteristic for C11-SiNCs that synthesized by this method as observed by our research group several years ago [26], [28], [45]-[47].

Fig. 4 UV-Vis absorption spectra of C11-SiNCs dispersed in deionized water (black line) and Er/C11-SiNCs dispersed in deionized water (red line)

Fig. 5 (a) Reflected light image of C11-SiNCs collected before (b) confocal luminescence spectrum image of dried C11-SiNCs deposited from dichloromethane on the glass coverslip (c) reflected light image of Er/C11-SiNCs collected prior to (d) confocal luminescence spectrum image of dried Er/C11-SiNCs deposited from aqueous solution on the glass coverslip. The laser wavelength was 488 nm and the scale displays the scattered intensity integrated over the Stokes shift range from 200 cm⁻¹ to 7000 cm⁻¹, which corresponds to a mixture of Raman signals and the luminescence. The scale bar of the reflected light images (a,c) is 10 µm and the scan size of the luminescence images (b,d) is 50 X 50 µm

Fig. 6 (b) shows the average Raman/luminescence spectrum that observed from erbium mixed alkylated SiNCs that drop coated from deionized water onto glass coverslip. The characteristic large and very broad Raman peak presented in this spectrum with a maximum intensity at around 3960 cm⁻¹ (605 nm) can be referred to orange luminescence signals from C11-SiNCs [26]-[28], [44]. It is also should be noted that this peak is a broader than that of alkylated SiNCs. In addition, the position of this peak shifts to a lower wavelength (blue shift) due to the oxidation of alkylated SiNCs in presence of erbium ions. As mentioned The luminescence peak of C11-SiNCs that prepared by the electrochemical method [26] most likely to be affected by the oxidation reaction between the SiNCs and the water. In addition, these oxidations states of alkylated SiNCs (SiO₂) are affected when they mixed with rare earth ion i.e.
Er$^{3+}$ as obtained from the core line Si$_2$p in XPS spectrum of Er/C$_{11}$-SiNCs, see Fig. 3 (a). XPS spectrum shows that the Er/C$_{11}$-SiQDs have sub oxides on the surface of the nanoparticles. This gives evidence that the blue shift in the luminescence peak of SiNCs in Er/C$_{11}$-SiNCs arises from states that related with surface oxide. Thus, the surface chemical composition must play an important role in the blue shift of the luminescence of alkylated SiNCs when they mixed with erbium ions. It can be seen also that the typical peak of crystalline Si at 515 cm$^{-1}$ does not appear in presence of erbium trichloride. This observation may be a result of overlap with the Si orange luminescence peak to become one broad peak. Furthermore, the sharp features at around 5145 cm$^{-1}$ likely due to the electronic transitions from $^4$F$^{9/2}$ to the ground state $^4$I$_{15/2}$ in erbium ions[48].

It also can be seen that the intensity of the scattered Raman light from the mixture is higher than that of pure alkylated SiNCs. This enhancement in the intensity can be explained as a result of the presence of erbium ions on the surface of alkyl capped SiNCs that influence the luminescence intensity of Er/C$_{11}$-SiNCs. Therefore, the typical absorption spectrum of erbium trichloride is carried out in order to interpret this significant improvement in the intensity of the PL peak of Er/C$_{11}$-SiNCs as shown in Fig. 7. This optical absorption spectrum of erbium trichloride solution shows various bands that correspond to different absorption transitions of erbium ions [48], [49]. These electronic transitions bands are listed in Table II. From the typical absorption spectrum of erbium trichloride (Fig. 7), there is an electronic transition peak at 488 nm which attributed to $^4$F$^{7/2}$ [49]. This transition band is corresponding to the laser wavelength at 488 nm of Raman spectroscopy that used to excite the Er/C$_{11}$-SiNCs sample. Consequently, it is most likely that the laser light has absorbed by Er$^{3+}$ in this transition band and the energy transferred to the C$_{11}$-SiNCs that reflect the significant enhancement in the intensity of the luminescence band of SiNCs when they mixed with erbium ions (see Fig. 6).

<table>
<thead>
<tr>
<th>Energy level</th>
<th>$\lambda_{max}$ (nm)</th>
</tr>
</thead>
<tbody>
<tr>
<td>$^4$F$^{9/2}$</td>
<td>652</td>
</tr>
<tr>
<td>$^4$S$^{3/2}$</td>
<td>545</td>
</tr>
<tr>
<td>$^2$H$^{11/2}$</td>
<td>521</td>
</tr>
<tr>
<td>$^4$F$^{7/2}$</td>
<td>488</td>
</tr>
<tr>
<td>$^4$F$^{5/2}$</td>
<td>449</td>
</tr>
<tr>
<td>$^4$F$^{3/2}$</td>
<td>442</td>
</tr>
<tr>
<td>$^2$H$^{9/2}$</td>
<td>405</td>
</tr>
<tr>
<td>$^4$G$^{11/2}$</td>
<td>379</td>
</tr>
<tr>
<td>$^2$G$^{9/2}$</td>
<td>364</td>
</tr>
<tr>
<td>$^2$G$^{7/2}$</td>
<td>356</td>
</tr>
<tr>
<td>$^4$D$^{7/2}$</td>
<td>255</td>
</tr>
</tbody>
</table>

Table II: The experimental absorption peaks of erbium energy level.
The PL emission spectrum of C\textsubscript{11}-SiNCs drop coated from deionized water onto glass coverslip is presented in Fig. 8 (the excitation wavelength = 400 nm). The sample reveals an orange emission peak at 595 nm (2.1 eV). This peak can be attributed to the band gap transition of alkyl capped SiNCs which is assigned to radiative recombination of excitons inside alkyl capped SiNCs [45]. It is known that the higher energy is required to generate the exciton luminescence in SiNCs which is higher than the band gap energy of bulk silicon due to quantum confinement [25].

XPS spectrum, the absorbance spectrum, Raman spectrum and the emission spectrum of alkyl capped SiNCs that studied in this work are identical with previous work which reported for alkylated SiNCs [26], [27], [29], [30]. Thus, these data indicate the successful synthesis of alkyl capped SiNCs.

Fig. 9 indicates the PL emission spectra of erbium mixed alkylated SiNCs which drop coated from deionized water onto glass coverslip. This sample also exhibits the characteristic C\textsubscript{11}-SiNCs PL band at 595 nm in the region between 500 and 750 nm as seen in Fig. 9 (a) (the excitation wavelength was 400 nm) [45]. The position of this Si peak (595 nm) does not change in the presence of erbium ions; while the intensity of this PL band is considerably higher than the PL intensity band of C\textsubscript{11}-SiNCs as presented in Fig. 10; where the concentrations of C\textsubscript{11}-SiNCs are equal in both samples i.e. 0.08 g/L. Fig. 9 (a) presents also PL band at 545 nm which assigned to 4S\textsubscript{3/2} of Er ion [49].

It can be seen also that the peak at 595 nm of erbium mixed alkyl capped SiNCs shows a fourfold increase in the PL emission intensity. This enhancement in the intensity is similar to that observed from Raman spectrum of Er/C\textsubscript{11}-SiNCs. As mentioned above, this can be explained as a result of the presence of erbium ions on the surface of alkyl capped SiNCs, which influence the luminescence intensity of Er/C\textsubscript{11}-SiNCs. The absorption spectrum of erbium shows an electronic transition peak at around 405 nm which is assigned to 2H\textsubscript{11/2} [49]. This band is very close to the laser wavelength at 400 nm that used to excite the Er/C\textsubscript{11}-SiNCs. Hence, it is quite possible that erbium absorbing laser light in this transition band and the energy transferred to the C\textsubscript{11}-SiNCs from the erbium ion causing a significant enhancement in the intensity of the PL emission band of SiNCs when they mixed with erbium ions.

The PL infrared emission peak at around \( \approx 1536 \) nm in the region between 1400 nm and 1700 nm attributed to the intra-4f transition of trivalent erbium (intra-atomic \( 4I_{13/2} \rightarrow 4I_{15/2} \) transitions of Er\textsuperscript{3+}) [22] and the excitation wavelength was 520 nm, see Fig. 9 (b). It is known that erbium has an important role in optical fiber technology due to its luminescence at around 1535 nm [1]. In addition, the main goal of mixing SiNCs with erbium is to observe the energy transferred from the excited state of Si to the erbium. Thus, Figs. 9 (a), (b) demonstrated that Er/C\textsubscript{11}-SiNCs sample exhibits simultaneously two PL emission bands corresponding to radiative recombination of electron-hole pairs in alkylated SiNCs (595 nm) and the intra-4f transition in Er\textsuperscript{3+} (1536 nm). The latter peak is very weak and the intensity at 1536 nm is around a factor of 1 which determines very low PL emission intensity from erbium ions. While, the Si PL peak becomes high in the intensity, as the erbium trichloride exists.

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**Fig. 8 PL emission spectrum of alkylated SiNCs drop coated from deionized water onto glass coverslip. The excitation wavelength = 400 nm**

**Fig. 9 PL emission spectra of erbium mixed alkylated SiNCs drop coated from deionized water onto glass coverslip showing two different regions (a) 500 – 750 nm and the excitation wavelength was 400 nm (b) 1400 – 1700 nm and the excitation wavelength was 520 nm**
The lack of the PL Er\(^{3+}\) intensity band (1536 nm) can be due to the mixing method used in this study, which does not grow the PL intensity of erbium in Er/C\(_{11}\)-SiNCs mixture, but it causes a significant enhancement in the PL peak of Si nanoparticles. Thus, there is no energy transfer from excited alkylated SiNCs to Er\(^{3+}\) as we expected [21], [50] while the energy transfer from erbium ions to alkylated SiNCs. In order to transfer energy from SiNCs to erbium ions, the enhancement of 1536 nm PL emission band should be achieved. Assuming that the energy transferred from alkylated SiNCs to erbium ions (Er\(^{3+}\)), Fig. 9 should be interpreted as follow. Basically, the photons are absorbed by alkylated SiNCs and then excitons are generated in the SiNCs [51]. A fraction of recombination energies of the excitons are transferred then to the trivalent erbium. The increased of the total transferred energy to erbium ions depends on the chemical environment. This generates enhancement in the PL intensity peak of erbium (1536 nm) and quenching in the PL intensity peak of Si (595 nm). While, the results from this chemical mixing method which present in Figs. 9 (a), (b) shows growth in the 595 nm PL peak of Si and quenching in the 1536 nm PL band of Er\(^{3+}\), which provides a clear evidence that the energy does not transfer from alkylated SiNCs to Er\(^{3+}\) as the concentration of SiNCs kept constant in both cases. This can be explained as a result of presence of the erbium in an oxide form i.e. Er\(_{2}O_{3}\) as confirmed from XPS results of Er mixed alkylated SiNCs. Thus, the presence of oxides between the erbium and silicon core forbids this energy transfer. Kik et al. [52] and Ji et al. [24] observed that the efficient enhancement in the PL band of erbium at \(1530\) nm occur when the Er\(^{3+}\) are spaced closely enough to SiNCs in order to allow the energy transfer process.

However, the chemical mixing route that used in this work does not provide efficient PL from erbium ions when they mixed with SiNCs. It is noteworthy that erbium ions have an important emission wavelength centered at about 1535 nm (characteristic for the intra-4\(^f\) transition of Er\(^{3+}\)) which correspond to a standard optical telecommunication wavelength [53]. Thus, this study aims to achieve maximum emission intensity from trivalent erbium which generated as a result of the energy transfer from excited silicon nanocrystals to the nearby erbium ion. Regarding to data of Er/C\(_{11}\)-SiNCs, this simple chemical mixing of SiNCs and erbium ions leads to transferring energy from erbium ions to SiNCs not as we expected [21], [50] from excited SiNCs to Er\(^{3+}\). Furthermore, this failure to observe optical active erbium may be due to the concentration of erbium that experimentally used which was not sufficient to get efficient Er emission from the samples but affects the photoluminescence of SiNCs considerably. In contrast, previous works [15], [21], [22], [25], [54] have been reported a significant increase in the intensity of the PL emission peak of erbium when the erbium doped silicon nanocrystals by different doping methods such as ion implantation method and sputtering procedure. It was observed that intensity of 1536 nm peak increased dramatically in the presence of silicon nanocrystals and the PL emission peak of silicon nanocrystals is quenched.

However, [21], [15], [54] found that the size of silicon nanocrystals, the chemical environment of erbium doped silicon nanocrystals films and different concentration of erbium ions inside the silicon nanocrystals film affect the PL emission intensities of Si and Er\(^{3+}\). Fujii et al. [21] and Cerqueira et al. [15], [54] found that the intensity of PL peak of erbium ions (1545 nm) increased rapidly with decreasing the size of SiNCs from 3.8 nm to 2.7 nm and from 8 nm to 3 nm, respectively. In addition, they found that the PL intensity of erbium ions increases as the concentration of erbium decreased within the SiNCs. Thus, different intensities of the PL emission bands can be observed depending on the above factors. Thus, it is apparent that the chemical mixing method, chemical environment of the erbium mixed SiNCs and the concentration of erbium trichloride that practically utilized (4 mM) does not transfer energy from SiNCs to Er ion. It is also exert an insignificant effect on the PL efficiency of erbium ions but there is a large effect on the luminescence of SiNCs.

IV. CONCLUSION

Alkylated silicon nanocrystals (C\(_{11}\)-SiNCs) were prepared successfully by galvanostatic etching of p-Si(100) wafers followed by a thermal hydrosilation reaction of 1-undecene in refluxing toluene in order to extract C\(_{11}\)-SiNCs from porous silicon. Erbium trichloride was added to alkylated SiNCs using a simple mixing chemical route. To the best of our knowledge, this is the first investigation on mixing SiNCs with erbium ions (III) by this chemical method. However, the main goal of mixing SiNCs with erbium ions is to observe the energy transfer from the excited state of SiNCs to erbium. Erbium is useful because it has a narrow emission spectrum centered at \(~1500\) nm which is a range of interest in optical fiber technology. The erbium trichloride hydrolysis to erbium sesquioxide (Er\(_{2}O_{3}\)) in Er/C\(_{11}\)-SiNCs film, which coat the surface of, alkylated SiNCs. The presence of Er\(_{2}O_{3}\) on the surface of C\(_{11}\)-SiNCs affects the oxidation state of alkylated
SiNCs on the surface as observed by XPS measurement. C_{11}-SiNCs and their mixtures with Er^{3+} were investigated using Raman spectroscopy and photoluminescence (PL). The characteristic broad luminescence peak confirmed the presence of SiNCs. The wavelength of this luminescence signal of SiNCs in aqueous medium in presence of trivalent erbium shifts to a lower wavelength due to the formation of oxides which would cause a blue shift to higher energy based on the quantum confinement effect. This suggests that the blue shift arises from states that related to the surface oxide on SiNCs. Thus, the surface chemical composition must play an important role in the blue shift of the luminescence of SiNCs when they mixed with erbium ions. This data suggests that the origin of the luminescence from C_{11}-SiNCs could be attributed to direct band gap transitions in SiNCs and to radiative recombinations of excitons through the oxidation surface of SiNCs. In addition, the Er/C_{11}-SiNCs sample shows an orange PL emission peak at around 595 nm originates from Si. Er/C_{11}-SiNCs mixture also exhibits a weak PL emission peak at 1536 nm that originates from the intra-4f transition in erbium ions (Er^{3+}). The PL peak of Si in Er/C_{11}-SiNCs mixtures increased in the intensity up to three times as compared to pure C_{11}-SiNCs. The collected data suggest that this chemical mixing route leads instead to a transfer of energy from erbium ions to SiNCs.

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